The thermal stability of AIN

A. ABID, R. BENSALEM, B. J. SEALY Department of Electronic and Electrical Engineering, University of Surrey, Guildford, Surrey, UK

The thermal stability of AIN powders and thin films has been investigated using reflection high-energy electron diffraction (RHEED) and X-ray diffraction. AIN powder was treated thermally and chemically to assess the oxidation resistance of this compound and to identify the phases formed. The results show that AIN is stable up to 1000° C in air and remains stable up to 1400° C *in vacuo*. γ -AIOOH is formed when AIN is treated with water at 100° C but AIN does not react readily with atmospheric moisture at room temperature. The thermal stability of thin films of AIN on GaAs has been evaluated at temperatures between 900 and 1100° C in a nitrogen atomosphere. It was found that AIN did not oxidize under these conditions. Pure AIN is a suitable encapsulant for GaAs at high annealing temperatures in an inert atmosphere.

1. Introdution

Aluminium nitride is currently of great interest as an encapsulant to protect ion-implanted GaAs during annealing. The chemcial reactivity of AlN under these conditions is thus of practical importance. Kubaschevski and Evans [1] have reported that AIN decomposes when heated in air at 940 to 950° C whereas Lyutaya et al. [2, 9] state that AIN oxidizes when heated in air above 800° C. In addition, the oxidation of crystals of AIN has been studied by Slack and Chelly [3] who found that at room temperature a protective oxide layer, about 10 nm thick, formed on single crystals of AlN in 1 day. Lavrenku et al. [4] showed that the oxidation rate of sintered AlN in oxygen at 800 to 1100° C was very low and that at 1100° C only 8% was oxidized. Thus it was thought necessary to investigate the stability of AlN under conditions likely to be encountered as a result of encapsulation and annealing of ion-implanted GaAs. X-ray diffraction (XRD), and reflection high energy electron diffraction (RHEED) have been used to study treated AlN powders and thin films were used to investigate bulk and surface phases formed.

2. Experimental details

AlN powder (99%) was supplied by Koch Light Limited (UK) and had a particle size of $50 \,\mu\text{m}$ and a dark grey colour, and by Alfa Products (West Germany) which had a light grey colour and a finer particle size. XRD analysis was carried out using a Debye–Scherrer camera of 11.486 mm diameter CuK α radiation and glass specimen capillaries of 0.5 mm diameter. The RHEED analysis was carried out using an operating voltage of 40 kV at background pressure of 3 to 5 × 10⁻⁵ torr. The sample holder was an aluminium block (2 cm × 1 cm × 1 cm) with a wide, shallow groove into which the powder was pressed. The surface was made as smooth as possible before analysis.

Samples of AlN powder contained in porcelain boats were heated in air in a muffle furnace for 3 h at

0022-2461/86 \$03.00 + .12 © 1986 Chapman and Hall Ltd.

temperatures from 600 to 1400° C. The samples were allowed to cool in the furnace before being analysed. Other samples of AlN powder were placed in quartz capsules 7 cm long and 2 cm diameter, which were evacuated to about 10^{-4} torr and sealed. These capsules were heated in a muffle furnace for different times at temperatures [3] from 600 to 1400° C, and then allowed to cool in the furnace.

In order to study the wet oxidation of AlN and to identify any oxidation products, AlN powder was treated with 30 wt % hydrogen peroxide. About 0.5 g AlN was placed in a 250 ml beaker with $125 \text{ ml H}_2\text{O}_2$ solution, covered with a clock glass and either (a) heated to boiling point for 10 min, or (b) stirred for 5 min and left at room temperature for 24 h. The suspensions were then filtered and the residues left to dry uncovered at room temperature for 24 h. The reaction of AlN with water was studied by placing about 0.5 g AlN into 125 ml water and boiling for 10 min, when ammonia gas was liberated and a white precipitate formed which was removed by filtration and then dried for 24 h in air. Thin films of AlN were preapared by vacuum evaporation of aluminium metal in an atmosphere of nitrogen and ammonia [5]. The annealing of AIN films was carried out using a graphite strip heater for a very short time (30 sec) in flowing nitrogen and at a temperature of 900 to 1100° C. This process is typical for the rapid thermal annealing of ion-implanted GaAs.

3. Results

Table I shows the results obtained for AlN heated in air for 3 h. The XRD analysis gives the composition of the bulk sample and the RHEED analysis gives that of the surface layer to a depth of a few nanometres. The typical RHEED pattern of the AlN powder at room temperature is shown in Fig. 1. For AlN heated to 1000° C the RHEED analysis revealed the presence of an unidentified phase (X, Fig. 2) which is possibly an oxide or oxynitride. However, on heating to 1200° C, the AlN was oxidized to α -Al₂O₃ (Fig. 3) but still

TABLE I Effect of heating AlN powder in air * – bulk and surface

Temp (° C)	Phases present after heating for 3 h				
	XRD analysis (bulk)	RHEED analysis (surface)			
600	AlN	AlN			
800	AIN	AIN			
1000	AIN	phase X			
1200	$\alpha - Al_2O_3$	phase X + α -Al ₂ O ₃			
1400	α -Al ₂ O ₃	α -Al ₂ O ₃			

*Annealing *in vacuo* showed no sign of oxidation even for samples heated to 1400° C for more than 24 h.

contained a second phase on the surface which had a strong unidentified diffraction line at 0.288 nm (see Table II).

The samples of AIN heated in vacuo in quartz capsules for 24 h at temperatures from 600 to $1400^{\circ}\,C$ contained only AlN when analysed by both XRD and RHEED. Table II shows the *d*-spacings of phase X obtained from the RHEED pattern of AlN heated in air at 1000° C and also shows the *d*-spacings obtained from AlN heated in air at 1200°C and in vacuo at 1400° C. Measured *d*-spacings of *γ*-AlOOH (Fig. 4) obtained from the surface of AIN reacted with boiling water, are also listed in the table and the standard patterns of AlN and a-Al₂O₃ are included for comparison. RHEED patterns from thin films of AlN on (100) GaAs are shown in Fig. 5. The diffraction pattern from the as-deposited AlN film (Fig. 5a) reveals the polycrystalline structure of the deposited layer and shows the presence of AlN only. This type of film, when annealed at 950° C, gave the RHEED pattern shown in Fig. 5b. Sharpening of the rings is clearly observed, but no change in the structure has occurred. An AlN thin film of the same type annealed at a higher temperature (1100 $^{\circ}$ C) gave the diffraction pattern shown in Fig. 5c. Again, the diffraction lines are those of AlN alone, clearly demonstrating the oxidation resistance of the AlN thin film. However, some of the rings are "spotty" and this phenomenon may be explained as being due to recrystallization occurring in the layer at this temperature.

4. Discussion

AlN powder proved to be resistant to oxidation in air



Figure 2 RHEED pattern for AlN powder heated at 1000° C.

at temperatures up to about 1000° C. At higher temperatures an unknown phase, X, was formed on the surface. This phase could be an oxide or oxynitride of aluminium. The unknown phase, X, and α -Al₂O₃ seem to be related structures, but phase X is not α -Al₂O₃ since the strongest line of α -Al₂O₃ (0.346 nm) is absent in phase X. Furthermore, phase X was formed when thin films of aluminium were annealed at about 1000° C in nitrogen containing traces of air. On heating to 1200° C in air, AlN was transformed completely into α -Al₂O₃, although the pattern from the surface still contains weak traces of the line at 0.288 nm. When heated to 1400° C in evacuated capsules (10^{-4} torr) AlN powder did not react at the low oxygen pressure prevailing.

Thin films of AlN deposited on GaAs and then annealed in the temperature range 900 to 1100° C in a nitrogen atmosphere showed no sign of oxidation. However, at higher temperatures some sharpening of the arcs occurred. The appearance of spots in the arcs indicated recrystallization of some of the AlN at 1100° C. The stability of AlN at lower temperatures under various conditions was also investigated. AlN did not react with the oxidizing agent hydrogen peroxide, again indicating the stability of AlN towards oxidation. However, AlN reacted with hot water and the surface was converted to AlOOH (boehmite). Polycrystalline AlN did not form a surface oxide in air at room temperature, although Slack and Chelly [3] have reported that powders and single crystals form a protective layer of oxide 5 to 10 nm thick. Such a crystalline oxide was not detected in the



Figure 1 Standard RHEED pattern taken from powder AlN at room temperature.



Figure 3 RHEED pattern obtained from AlN heated at 1200°C.

Standards			Results								
Standard RHEED pattern of AlN, room temp.		Standard RHEED pattern α -Al ₂ O ₃ ,		AlN heated to 1400° C in vacuo		AlN heated in air at 1000° C		AlN heated at 1200° C in air		AlN reacted with H_2O at 100° C	
		room temp.		<i>d</i> (nm)	I	d (nm)	1	d (nm)	1	<i>d</i> (nm)	I
<i>d</i> (nm)	<i>I</i> *	<i>d</i> (nm)	Ι							, ·	
		0.346	s					0.346	s		
						0.200	0	0.200		0.317	S
0 272	VS			0.272	VS	0.290	3	0.200	8		
0.250	m	0.252	s	0.250	m	0.250	vs	0.250	s		
0.238	vs	0.202	5	0.238	vs	0.200	10	0.250	5	0.235	s
		0.207	s			0.205	s	0.208	s	0.255	5
0.183	s			0.184	s					0.188	vs
		0.175	m					0.176	m		
		0.159	m					0.159	m	0.165	m
0.155	s			0.155	s	0.155	w				
0.144	s	0.141	vs	0.141	S	0.144	vs	0.140	s	0.144	vs
0.131	s			0.131	s					0.132	vs
						0.122	w	0.122	w		
0.118	m			0.118	m						
0.115	m			0.115	m						
Commonts										0.113	m
Lomments		In good		Good		Unknown		Mainly			
in good		agreement		agreement						y-Aloon boshmita	
with ASTM		ugith ASTM		with AIN		formed		$u - Al_2 O_3$,		formed	
data card		data card		at room		Tormeu		unknown		Tormed	
no. 25-1133		no 10-173		temperature				unknown			

TABLE II Lattice dspacings (nm) from RHEED patterns from surface of oxidized AIN

*vs = very strong; s = strong; m = medium; w = weak.

present work. A crystalline oxide layer would have given its own RHEED pattern and an amorphous oxide layer would have blurred the RHEED pattern. The aluminium oxide hydroxide pattern obtained from the reaction with water showed that AlN does not dissolve readily in water to form aluminium hydroxide. The formation of the crystalline layer of AlOOH prevented further reaction of the AlN with water. It had previously been reported that treatment with water gave either Al(OH)₃ [6]:

$$AlN + 3H_2O \rightarrow NH_3 + Al(OH)_3$$

or α-Al₂O₃ [7]

$$2AIN(s) + 3H_2O(g) \rightarrow NH_3(g) + \alpha - AI_2O_3(s)$$

The present work shows that the probable reaction is:

 $AlN + 2H_2O \xrightarrow{100^\circ C} NH_3 + AlOOH$

Figure 4 RHEED pattern obtained from AlN reacted with hot H_2O (γ -AlOOH).

5. Conclusions

AlN is suitable as an encapsulant for GaAs under non-oxidizing conditions up to 1400° C; this is at least 200° C higher than might be required in practice for annealing GaAs. Some recrystallization of the AlN thin film occurred between 950 and 1100° C. Polycrystalline AlN is also stable at room temperature under normal conditions of use.

Acknowledgements

The authors would like to thank Mrs P. Bills and Dr D. Lewis from the Chemistry Department for their useful comments. They also would like to thank Mr S. Greaves for carrying out the X-ray analysis.



Figure 5 RHEED patterns obtained from thin AlN films on (100) GaAs. (a) As-deposited by vacuum evaporation of pure Al metal in N_2 -NH₃ mixture at 3 × 10⁻³ torr. (b) Annealed at 950°C for 30 sec. (c) Annealed at 1100°C for 30 sec.



Figure 5 Continued.

References

- 1. O. KUBASCHEVSKI and E. L. EVANS, Met. Thermochem. (Russian Trans) 11 (1954) 254.
- 2. G. V. SAMSONOV and I. M. VINITSKII (eds), in "Handbook of refractory Compounds", (1980) p. 386.
- 3. G. A. SLACK and T. F. M. CHELLY, J. Cryst. Growth 34 (1976) 263.
- 4. V. A. LAVRENKU, A. F. ALEKSEEV, F. S. LUYOV-SKAYA and I. N. FRANTESEVICH, *Dokl. Akad. Nauk. SSSR* 255 (1980) 641 (in Russian) 0002-3264 (*Metal Abstract* 1981, 34-1281).
- 5. R. BENSALEM, N. J. BARRETT and B. J. SEALY, *Electron. Lett.* 19 (1983) 112.



- 6. R. B. HESLOP and K. JONES, in "Inorganic Chemistry" (Elsevier, Amsterdam, 1976) p. 341.
- 7. Y. PAULEAU, A. BOUTEVILLE, J. J. HANTZPERGUE, J. C. REMY and A. CACHARD, J. Elect. Chem. Soc. Solid State Sci. Tech. 129 (1982) 1045.
- 8. R. BENSALEM, A. ABID and B. J. SEALY, *Thin Solid Films*, to be published.
- 9. M. D. LYUTAYA and V. F. BUKHANEVICH, Zh. Neorg. Khim. 7(11) (1962) 2487.

Received 9 May and accepted 12 June 1985