The thermal stability of AIN

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The thermal stability of AIN powders and thin films has been investigated using reflection high-energy electron diffraction (RHEED) and X-ray diffraction. AIN powder was treated thermally and chemically to assess the oxidation resistance of this compound and to identify the phases formed. The results show that AIN is stable up to 1000°C in air and remains stable up to 1400°C *in vacuo.* y-AIOOH is formed when AIN is treated with water at 100°C but AIN does not react readily with atmospheric moisture at room temperature. The thermal stability of thin films of AIN on GaAs has been evaluated at temperatures between 900 and 1100°C in a nitrogen atomosphere. It was found that AIN did not oxidize under these conditions. Pure AIN is a suitable encapsulant for GaAs at high annealing temperatures in an inert atmosphere.

1. Introdution

Aluminium nitride is currently of great interest as an encapsulant to protect ion-implanted GaAs during annealing. The chemcial reactivity of A1N under these conditions is thus of practical importance. Kubaschevski and Evans [1] have reported that A1N decomposes when heated in air at 940 to 950° C whereas Lyutaya et al. [2, 9] state that AlN oxidizes when heated in air above 800° C. In addition, the oxidation of crystals of A1N has been studied by Slack and Chelly [3] who found that at room temperature a protective oxide layer, about 10nm thick, formed on single crystals of A1N in 1 day. Lavrenku *et al.* [4] showed that the oxidation rate of sintered A1N in oxygen at 800 to 1100° C was very low and that at 1100° C only 8% was oxidized. Thus it was thought necessary to investigate the stability of A1N under conditions likely to be encountered as a result of encapsulation and annealing of ion-implanted GaAs. X-ray diffraction (XRD), and reflection high energy electron diffraction (RHEED) have been used to study treated A1N powders and thin films were used to investigate bulk and surface phases formed.

2. Experimental details

AIN powder (99%) was supplied by Koch Light Limited (UK) and had a particle size of 50 μ m and a dark grey colour, and by Alfa Products (West Germany) which had a light grey colour and a finer particle size. XRD analysis was carried out using a Debye-Scherrer camera of 11.486mm diameter CuK α radiation and glass specimen capillaries of 0.5 mm diameter. The RHEED analysis was carried out using an operating voltage of 40 kV at background pressure of 3 to 5 \times 10⁻⁵ torr. The sample holder was an aluminium block (2 cm \times 1 cm \times 1 cm) with a wide, shallow groove into which the powder was pressed. The surface was made as smooth as possible before analysis.

Samples of A1N powder contained in porcelain boats were heated in air in a muffle furnace for 3 h at

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temperatures from 600 to 1400° C. The samples were allowed to cool in the furnace before being analysed. Other samples of A1N powder were placed in quartz capsules 7cm long and 2cm diameter, which were evacuated to about 10^{-4} torr and sealed. These capsules were heated in a muffle furnace for different times at temperatures [3] from 600 to 1400° C, and then allowed to cool in the furnace.

In order to study the wet oxidation of A1N and to identify any oxidation products, A1N powder was treated with 30 wt % hydrogen peroxide. About 0.5 g AlN was placed in a 250 ml beaker with 125 ml $H₂O₂$ solution, covered with a clock glass and either (a) heated to boiling point for 10min, or (b) stirred for 5min and left at room temperature for 24h. The suspensions were then filtered and the residues left to dry uncovered at room temperature for 24h. The reaction of A1N with water was studied by placing about 0.5g A1N into 125ml water and boiling for 10min, when ammonia gas was liberated and a white precipitate formed which was removed by filtration and then dried for 24 h in air. Thin films of A1N were preapared by vacuum evaporation of aluminium metal in an atmosphere of nitrogen and ammonia [5]. The annealing of A1N films was carried out using a graphite strip heater for a very short time (30 sec) in flowing nitrogen and at a temperature of 900 to 1100° C. This process is typical for the rapid thermal annealing of ion-implanted GaAs.

3. Results

Table I shows the results obtained for A1N heated in air for 3 h. The XRD analysis gives the composition of the bulk sample and the RHEED analysis gives that of the surface layer to a depth of a few nanometres. The typical RHEED pattern of the A1N powder at room temperature is shown in Fig. 1. For A1N heated to 1000° C the RHEED analysis revealed the presence of an unidentified phase $(X, Fig. 2)$ which is possibly an oxide or oxynitride. However, on heating to 1200° C, the AIN was oxidized to α -Al₂O₃ (Fig. 3) but still

TABLE I Effect of heating AlN powder in air* $-$ bulk and surface

Temp $(^{\circ}C)$ 600	Phases present after heating for 3h					
	XRD analysis (bulk)	RHEED analysis (surface)				
	AIN	AIN				
800	AIN	A1N				
1000	AlN	phase X				
1200	α -Al ₂ O ₃	phase $X + \alpha - Al_2O_3$				
1400	α -Al ₂ O ₃	α -Al ₂ O ₂				

*Annealing *in vaeuo* showed no sign of oxidation even for samples heated to 1400°C for more than 24h.

contained a second phase on the surface which had a strong unidentified diffraction line at 0.288 nm (see Table II).

The samples of AIN heated *in vacuo* in quartz capsules for 24h at temperatures from 600 to 1400° C contained only A1N when analysed by both XRD and RHEED. Table II shows the d -spacings of phase X obtained from the RHEED pattern of A1N heated in air at 1000° C and also shows the *d*-spacings obtained from AlN heated in air at 1200°C and *in vacuo* at 1400 $^{\circ}$ C. Measured *d*-spacings of γ -AlOOH (Fig. 4) obtained from the surface of A1N reacted with boiling water, are also listed in the table and the standard patterns of AlN and α -Al₂O₃ are included for comparison. RHEED patterns from thin films of A1N on (100) GaAs are shown in Fig. 5. The diffraction pattern from the as-deposited A1N film (Fig. 5a) reveals the polycrystalline structure of the deposited layer and shows the presence of A1N only. This type of film, when annealed at 950° C, gave the RHEED pattern shown in Fig. 5b. Sharpening of the rings is clearly observed, but no change in the structure has occurred. An AIN thin film of the same type annealed at a higher temperature (1100 $^{\circ}$ C) gave the diffraction pattern shown in Fig. 5c. Again, the diffraction lines are those of A1N alone, clearly demonstrating the oxidation resistance of the A1N thin film. However, some of the rings are "spotty" and this phenomenon may be explained as being due to recrystallization occurring in the layer at this temperature.

4. Discussion

A1N powder proved to be resistant to oxidation in air

Figure 2 RHEED pattern for AlN powder heated at 1000°C.

at temperatures up to about 1000° C. At higher temperatures an unknown phase, X, was formed on the surface. This phase could be an oxide or oxynitride of aluminium. The unknown phase, X, and α -Al₂O₃ seem to be related structures, but phase X is not α -Al₂O₃ since the strongest line of α -Al₂O₃ (0.346 nm) is absent in phase X. Furthermore, phase X was formed when thin films of aluminium were annealed at about 1000° C in nitrogen containing traces of air. On heating to 1200° C in air, AlN was transformed completely into α -Al₂O₃, although the pattern from the surface still contains weak traces of the line at 0.288 nm. When heated to 1400° C in evacuated capsules (10^{-4} torr) A1N powder did not react at the low oxygen pressure prevailing.

Thin films of A1N deposited on GaAs and then annealed in the temperature range 900 to 1100° C in a nitrogen atmosphere showed no sign of oxidation. However, at higher temperatures some sharpening of the arcs occurred. The appearance of spots in the arcs indicated recrystallization of some of the A1N at 1100° C. The stability of AlN at lower temperatures under various conditions was also investigated. A1N did not react with the oxidizing agent hydrogen peroxide, again indicating the stability of A1N towards oxidation. However, A1N reacted with hot water and the surface was converted to A1OOH (boehmite). Polycrystalline A1N did not form a surface oxide in air at room temperature, although Slack and Chelly [3] have reported that powders and single crystals form a protective layer of oxide 5 to 10nm thick. Such a crystalline oxide was not detected in the

Figure 1 Standard RHEED pattern taken from powder A1N at room temperature. **Figure 3 RHEED** pattern obtained from AIN heated at 1200°C.

Standards			Results								
Standard RHEED pattern of AIN, room temp.		Standard RHEED pattern α -Al ₂ O ₃ ,		AlN heated to 1400° C in vacuo		AlN heated in air at 1000° C		AlN heated at 1200° C in air		AIN reacted with $H2O$ at 100° C	
		room temp.		d (nm)	Ι	d (nm)	1	d (nm)	\boldsymbol{I}	d (nm)	Ι
d (nm)	I^*	d (nm)	Ι								
		0.346	${\bf S}$					0.346	\mathbf{s}		
						0.290	s	0.288		0.317	s
0.272	VS			0.272	VS				${\bf S}$		
0.250	m	0.252	${\bf s}$	0.250	m	0.250	VS	0.250	${\bf S}$		
0.238	VS			0.238	VS					0.235	s
		0.207	${\bf S}$			0.205	s	0.208	${\bf s}$		
0.183	${\bf S}$			0.184	${\bf S}$					0.188	VS
		0.175	m					0.176	m		
		0.159	m					0.159	${\bf m}$	0.165	m
0.155	s			0.155	${\bf S}$	0.155	W				
0.144	s	0.141	VS	0.141	s	0.144	VS	0.140	S	0.144	VS
0.131	s			0.131	s					0.132	$_{\rm VS}$
						0.122	W	0.122	W		
0.118	m			0.118	m						
0.115	m			0.115	m						
Comments										0.113	m
				Good		Unknown					
In good		In good						Mainly		γ -AlOOH boehmite	
agreement with ASTM		agreement with ASTM		agreement with AlN		phase X formed		α -Al ₂ O ₃ ,			
data card								some		formed	
no. 25-1133		data card no. 10-173		at room temperature				unknown			

TABLE II Lattice d-spacings (nm) from RHEED patterns from surface of oxidized AIN

*vs = very strong; s = strong; m = medium; $w = weak$.

present work. A crystalline oxide layer would have given its own RHEED pattern and an amorphous oxide layer woudl have blurred the RHEED pattern. The aluminium oxide hydroxide pattern obtained from the reaction with water showed that A1N does not dissolve readily in water to form aluminium hydroxide. The formation of the crystalline layer of A1OOH prevented further reaction of the A1N with water. It had previously been reported that treatment with water gave either $AI(OH)$ ₃ [6]:

$$
AlN + 3H2O \rightarrow NH3 + Al(OH)3
$$

or α -Al₂O₃ [7]

$$
2\text{AlN}(s) + 3\text{H}_2\text{O}(g) \rightarrow \text{NH}_3(g) + \alpha \cdot \text{Al}_2\text{O}_3(s)
$$

The present work shows that the probable reaction is:

AlN + 2H₂O $\frac{100^{\circ}C}{\circ}$ NH₃ + AlOOH

Figure 4 RHEED pattern obtained from AIN reacted with hot H_2O $(\gamma$ -AlOOH).

5. Conclusions

A1N is suitable as an encapsulant for GaAs under non-oxidizing conditions up to 1400 ~ C; this is at least 200~ higher than might be required in practice for annealing GaAs. Some recrystallization of the A1N thin film occurred between 950 and 1100°C. Poly**crystalline A1N is also stable at room temperature under normal conditions of use.**

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Figure 5 RHEED patterns obtained from thin A1N films on (1 00) GaAs. (a) As-deposited by vacuum evaporation of pure A1 metal in N_2-NH_3 mixture at 3×10^{-3} torr. (b) Annealed at 950°C for 30 sec. (c) Annealed at 1100° C for 30 sec.

Figure 5 Continued.

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